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Obtaining the scattering rate of different \( T_{\text{c0}} \) FeSe thin films via spectroscopic ellipsometry

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Due to the simplest crystalline structure among Fe-based superconductors, the FeSe system has attracted a lot of attention. In this work, FeSe thin films grown on the CaF\(_2\) substrate with \( T_{\text{c0}} = 6 \) and 11 K (named FeSe\(_1\) and FeSe\(_2\), respectively) are fabricated by a pulsed laser deposition technique. X-ray diffraction exhibits a high-quality single crystal of the two FeSe samples, and the lattice constants are about 5.574 Å. Atomic force microscopy characterizes their surface topography and roughness, which shows stripes in their surfaces that is helpful to construct a roughness layer using the optical measurement spectroscopic ellipsometry (SE) technique. SE is a powerful tool to determine FeSe thin films’ complex refractive index \( N = n + ik \) and plasma oscillation frequency \( \omega_p \). These important parameters are related to scattering rate \( \tau^{-1} \) for FeSe thin films. The results show that scattering rate \( \tau^{-1} \) of FeSe\(_2\) is significantly lower than that of FeSe\(_1\) in the whole frequency testing range at room temperature, which may be the reason that FeSe\(_2\) owns higher \( T_{\text{c0}} \) in low temperature than FeSe\(_1\). Published by the AVS.

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I. INTRODUCTION

High-temperature superconductors (HTSs) are the perspective materials in the applications of lossless power transmission, magnetic levitation transport, and magnetic resonance imaging.1 Since 2008, the discovery of the iron-based HTS has triggered much interest in these materials.2,3 In particular, the PbO-type FeSe system owing to its simplest crystalline structure has been one of the most studied ones among iron-based HTSs; thus, FeSe provides a unique opportunity to research its structural, electromagnetic transport, and optical properties.4–7 Among these properties, a blank space to research FeSe optical constants remains, which can help researchers to learn more about FeSe.

As we know, complex optical constants \( N = n + ik \) (\( n \) is the refractive index and \( k \) is the extinction coefficient) or complex dielectric functions \( \varepsilon = \varepsilon_r + i\varepsilon_i \) (\( \varepsilon_r \) and \( \varepsilon_i \) are the real part and the imaginary part of \( \varepsilon \), respectively) contain rich physical information. Because optical constants can not only characterize energy band transition for kinds of materials, for instance, two-dimensional (2D) MoS\(_2\),8 perovskite CsPbBe\(_3\)9 material and spinel oxide MgTi\(_2\)O\(_4\) (Ref. 10), etc., also are related with electrical quantities, such as scattering rate \( \tau \) and complex optical conductivity \( \sigma = \sigma_1 + i\sigma_2 \) (\( \sigma_1 \) and \( \sigma_2 \) are the real part and the imaginary part of \( \sigma \), respectively). Compared with reflectivity, SE measures the amplitude and phase information simultaneously and need not use Kramers–Kronig (K–K) transformation in a broad frequency testing range. Therefore, SE has become a suitable and convenient method to determine \( N \) or \( \varepsilon \).

As mentioned above, \( N \) or \( \varepsilon \) is associated with scattering rate \( \tau^{-1} \) and complex optical conductivity \( \sigma \), which are important parameters to analyze the intrinsic nature of FeSe. For example, Chinotti et al. provided that the interplay of the anisotropic scattering rate and Drude weight results in the nematic phase in the low energy excitation spectrum in the FeSe system.14 Yuan et al. measured the in-plane optical conductivity \( \sigma_1 \) of FeSe thin films and found two Drude components existing in the low-frequency optical conductivity \( \sigma_1 \) spectrum.7 Actually, it is worth noting that the plasma oscillation frequency \( \omega_p \) plays an important role in obtaining the scattering rate. In previous works, \( \omega_p \) is usually calculated by an integral form, whereas the upper limit of the integral is hard to decide.15 However, significant advantage exists in SE method for obtaining \( \omega_p \) because Drude dispersion law encompasses this parameter in fitting measurement data process.

Therefore, in this work, SE is employed to obtain complex refractive index \( N \) for two FeSe thin films with superconducting transition temperature \( T_{\text{c0}} = 6 \) and 11 K that are named FeSe\(_1\) and FeSe\(_2\), respectively. Drude and Lorentz oscillation dispersion laws are used to invert the SE measurement data \( \tan \Psi \) and \( \cos \Delta \), and \( \omega_p \) is derived using the Drude model. Finally, scattering rate \( \tau^{-1} \) is calculated through \( \sigma \) for FeSe\(_1\) and FeSe\(_2\), which may provide some information about the phenomenon in low temperature.

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II. EXPERIMENT

There are two FeSe thin films with $T_c = 6$ and $11$ K grown on CaF$_2$ substrates in this work, and they were deposited by the pulsed laser deposition (PLD) technique in a vacuum chamber. The details (synthesis and electrical measurement) of FeSe samples have been documented elsewhere. Crystallinity is characterized by x-ray diffraction (XRD, D8ADVANCE) with an x-ray wavelength of $\lambda = 1.54$ Å. Surface topography and roughness are measured by atomic force microscopy (AFM, Nanosurf). SE (GES-5, SOPRA) with a rotating analyzer is utilized to determine the complex optical constants of FeSe thin films, and the software WINELLI supplied by SOPRA is used to data analysis. To get smoother ellipsometric data, the spectral resolution is set to 5 nm in 300–800 nm wavelength (energy ranges from 1.55 to 4.13 eV and frequency from 12 500 to 33 333 cm$^{-1}$). The incident angle is fixed at 75°, which is close to Brewster angle and the testing condition is room temperature.

A. Crystal structure

As mentioned above, PbO-type FeSe has the simplest lattice structure in Fe-based superconductors that the FeSe unit consists of three layers with the central Fe layer sandwiched in between two adjacent Se layers, as shown in the inset of Fig. 1. The lattice diffraction patterns of FeSe$_1$ and FeSe$_2$ thin films grown on the CaF$_2$ substrate are exhibited in Fig. 1 by XRD. The two samples’ diffraction peaks show an orientation along (001), which demonstrates that FeSe thin films are fabricated with a high-quality single crystal. The c-axis lattice constant is calculated from the XRD results using the Bragg law,

$$2 \cdot d \cdot \sin \theta = n \cdot \lambda,$$

where $\theta$ is the diffraction angle and $n$ is the order of reflection. FeSe$_1$ and FeSe$_2$ exhibit similar values of the c-axis lattice parameter that is 5.574 Å, which is very close to the result of 5.8 Å reported by Feng et al. The high-quality samples open a credible door for the investigation of optical measurement and physical discussion.

B. Surface topography

AFM is not only a versatile tool to obtain abundant information on surfaces, but also a useful approach to get roughness data for removing this roughness layer to optical constants in SE analysis. Figure 2 is the surface condition of FeSe thin films. In the $2 \times 2 \mu$m$^2$ test range, homogenous light stripes are observed in the surface. The width of these stripes is about 208.5 and 260.1 nm for FeSe$_1$ and 2 samples that are illustrated using the pink arrows between double green lines in Figs. 2(b) and 2(e), respectively. The cross-section information is shown in Figs. 2(c) and 2(f), which can justifiably offer the visual roughness layer in building an optical structure, and this will be discussed in Sec. II C. Here, we do not focus the question on how these light stripes are formed, just put the point to SE analysis.

C. SE data analysis

1. Ellipsometry and experiment

As a nondestructive and accurate optical technique, $\tan \Psi$ and $\cos \Delta$ are the direct measurement values by SE, which are related to the amplitude ratio and phase difference between p- and s-polarizations, defined by the following equation:

$$\rho = \tan(\Psi)\exp(i\Delta) = \frac{r_p}{r_s},$$

where $r_p$ and $r_s$ express Fresnel coefficients for p- and s-polarized light, respectively. The Fresnel coefficients depend on the complex optical constants $N = n + ik$ and the thin film thickness. However, Eq. (2) is a nonlinear transcendental equation, and there is no possibility to directly obtain the analytic solutions. Generally, to achieve the complex optical constants by SE, one should construct the optical structure model that includes substrate, interface, film, and/or roughness layers information, etc., and film thickness to fit $\tan \Psi$ and $\cos \Delta$. Specially, one should emphatically take considering to the appropriate dispersion law and suitable free parameters for the layer of thin film that is measured because it is a crucial step to inverse measurement data.

2. Model construction

With the accurate and precision measurement values of $\tan \Psi$ and $\cos \Delta$, a suitable model including the optical model and the dispersion model should be constructed. First, to remove the effect of roughness on the complex optical constants or dielectric functions, the roughness layer should be added in the optical model. Thus, a four-phase optical model is used to describe the FeSe$_1$ and FeSe$_2$ film system, that is the void/roughness layer/FeSe layer/CaF$_2$ substrate. Based on AFM, the homogeneous triangle microstructure has been determined for FeSe$_1$; thus, the roughness...
layer consisted of 50% void and 50% FeSe described by the Bruggeman effective-medium approximation, and then, we fixed the concentration during the fitting process. In addition, the root mean square roughness is 5.44 nm by AFM and is 5.72 nm by SE, demonstrating that the roughness obtained by AFM can provide indicative information about SE data analysis. However, for FeSe$_2$, there are more burrs in the cross-section. Therefore, a good fitting result is 18% void.

Fig. 2. AFM surface characterization of (a) for FeSe$_1$ and (d) for FeSe$_2$. (b) and (e) illustrate the width of the light stripes, and (c) and (f) are the cross-section topography for the green arrows in (a) and (d) for FeSe$_1$ and FeSe$_2$ sample, respectively.

Fig. 3. Experimental data (marked by hollow circle) and simulated data (red lines for FeSe$_1$ and blue lines FeSe$_2$) of tan $\Psi$ and cos $\Delta$. 
and 82% FeSe_2 for the mixing concentration, and the thickness of roughness is 18.2 nm by SE, which is higher than that of 7.1 nm by AFM.

Then, Drude, describing the free electrons in metals, and Lorentz dispersion laws are chosen to fit $\tan \Psi$ and $\cos \Delta$ since FeSe is a superconductor in low temperature. A consistent fitting result between experimental and simulated data has been obtained, as shown in Fig. 3. Here, we simply introduce Drude and Lorentz dispersion law functions,

\[
\text{Drude: } \varepsilon_i = \frac{P}{\omega_p^2 + \omega^2} + \frac{\tau^{-1}}{\omega_p^2 + \omega^2},
\]

\[
\text{Lorentz: } \varepsilon_i = \frac{A\lambda^2 + (\lambda^2 - L_0^2)/[(\lambda^2 - L_0^2)^2 + \gamma^2\lambda^2]}{2},
\]

where $P$ is the polarization, $\omega_p$ is the plasma oscillation frequency, and $\tau^{-1}$ is the scattering rate of free electrons for the Drude model. For the Lorentz oscillator, $A$ is the intensity, $L_0$ is the central wavelength, and $\gamma$ is the width of the peak. Tables I and II list the best fitting parameters’ value for FeSe_1 and 2, respectively. The complex refractive index can be solved from the dielectric functions according to the following equations:

\[
\frac{\sigma}{n} = \sqrt{\frac{1}{2},}
\]

\[
\frac{k}{n} = \sqrt{\frac{1}{2},}
\]

3. Ellipsometry data analysis

After establishing the above optical model and dielectric functions, the theoretical results of $\tan \Psi$ and $\cos \Delta$ for FeSe thin films can be calculated by a regression method. To evaluate the goodness of the fitting parameters between experimental results and theoretical calculation, the mean squared error (MSE) is defined by the following function, which relative small values are exhibited in Table III, where $n$ is the number of $\tan \Psi$ and $\cos \Delta$, $m$ is the number of fitting parameters, and the subscripts cal and exp represent the theoretical calculation results and experimental results, respectively.

As we know, the dispersion law is an efficient method to invert the measurement data $\tan \Psi$ and $\cos \Delta$. However, point-by-point is also a resultful way to generate refractive index $n$ and extinction coefficient $k$.[17,20,21] Figure 4 is the n and k comparative results for the two FeSe samples by the methods of dispersion law marked by solid lines and point-by-point illustrated by gray lines. Very consistent consequences of $n$ and $k$ are obtained by the two means, indicating the validity of the Drude and Lorentz dispersion law and fitting parameters.

### III. RESULTS AND DISCUSSION

To understand why FeSe owns different $T_{c0}$ at low temperature, the scattering rate of free electrons $\tau^{-1}$ at room temperature may provide some information. $\tau^{-1}$ is related to optical conductivity $\sigma(\omega)$, whereas $\sigma(\omega)$ can be achieved by complex refractive index $N = n(\omega) + ik(\omega)$ or dielectric functions $\varepsilon = \varepsilon_r(\omega) + i\varepsilon_i(\omega)$. The equations are shown as follows:

\[
\tau^{-1}(\omega) = \frac{\omega_p^2}{4\pi} \frac{\sigma_1(\omega)}{\sigma_1^2(\omega) + \sigma_2^2(\omega)},
\]

\[
\sigma_1(\omega) = \frac{\omega[\varepsilon_r(\omega) + 1]}{4\pi} = \frac{\omega[2n(\omega)k(\omega) + 1]}{4\pi},
\]

| Table I. Fitting parameters’ value for the FeSe_1 sample. |
|---|---|---|---|
| $P$ | $\omega_p$ (nm$^{-1}$) | $\tau^{-1}$ (nm$^{-1}$) | Peak |
| 0.626 ± 0.089 | 12.611 ± 0.179 | 15.185 ± 0.216 | Peak1 |
| 2.187 ± 0.311 | 0.832 ± 0.012 | 0.813 ± 0.016 |
| 0.541 ± 0.007 | 0.220 ± 0.003 | 0.001 ± 0.001 |
| 0.165 ± 0.002 | 0.323 ± 0.004 | 0.101 ± 0.001 |

| Table II. Fitting parameters’ value for FeSe_2 sample. |
|---|---|---|---|
| $P$ | $\omega_p$ (nm$^{-1}$) | $\tau^{-1}$ (nm$^{-1}$) | Peak |
| 0.171 ± 0.002 | 7.521 ± 0.107 | 4.524 ± 0.064 | Peak1 |
| 0.688 ± 0.009 | 0.329 ± 0.005 | 0.140 ± 0.002 |
| 0.224 ± 0.003 | 0.715 ± 0.012 | 0.220 ± 0.003 |
| 0.746 ± 0.011 | 0.256 ± 0.003 | 0.009 ± 0.001 |
where $\omega_p$ is the plasma oscillation frequency which has been obtained by the Drude dispersion law in the fitting process in Eq. (8) and $\omega$ is the frequency.

Owing to the fact that $N = n(\omega) + ik(\omega)$ or $\varepsilon = \varepsilon_r(\omega) + i\varepsilon_i(\omega)$ have been characterized by SE, the optical conductivity $\sigma(\omega)$ is easy to be obtained. $\omega_p$ is derived by the Drude dispersion law, which are 12.611 and 7.521 $\mu$m$^{-1}$ for FeSe$_1$ and FeSe$_2$, respectively. These values are similar to the 10.9 $\mu$m$^{-1}$ characterized by optical reflectance $R(\omega)$ at room temperature on FeSe thin films grown on the SrTiO$_3$ substrate by Yuan et al.$^5$ Then, we substitute $\sigma(\omega)$ and $\omega_p$ into Eq. (8) and $\tau^{-1}/C_0$ could be obtained, which are illustrated in Fig. 5. After normalization, it is intuitive to conclude that $\tau^{-1}$ of FeSe$_2$ is obviously lower than that of FeSe$_1$ in the whole measurement frequency, which demonstrates that the metallicity of FeSe$_2$ is obviously better than FeSe$_1$. Therefore, we conclude that the higher $T_c$ in the FeSe$_2$ sample with 11 K maybe originated from the lower scattering rate compared with FeSe$_1$ with 6 K at room temperature.

The real part of optical conductivity $\sigma_1(\omega)$ is also shown in the inset of Fig. 5, the trend of frequency-dependent behavior is similar to the work of Wen et al.$^{22}$ as well as the FeSe counterpart Fe-pnictogen compounds.$^{23,24}$ It is worth noting that the peak at about 1.75 eV in FeSe$_2$ can be attributed to the transition from Fe $d$ level to Fe-$d$/Se-$p$ hybridized orbitals, corresponding to the two major peaks of the density of states near EF.$^{25}$ However, this transition has not been observed in FeSe$_1$, and the reason will be discussed in further.

### Table III. MSE results for FeSe$_1$ and FeSe$_2$.

<table>
<thead>
<tr>
<th></th>
<th>FeSe$_1$</th>
<th>FeSe$_2$</th>
</tr>
</thead>
<tbody>
<tr>
<td>MSE ($10^{-4}$)</td>
<td>3.298</td>
<td>3.131</td>
</tr>
</tbody>
</table>

![Fig. 4](https://example.com/fig4.png)

**Fig. 4.** Comparative results by the methods of dispersion law (solid lines) and point-by-point (gray lines). (a) and (b) are refractive index $n$ and extinction coefficient $k$ for FeSe$_1$, and (c) and (d) are $n$ and $k$ for the FeSe$_2$ sample.

![Fig. 5](https://example.com/fig5.png)

**Fig. 5.** Scattering rate of free electrons $\tau^{-1}$ depends on the frequency $\omega$ at room temperature for FeSe$_1$ (red line) and FeSe$_2$ (blue line). The inset is the real part of optical conductivity $\sigma_1(\omega)$ and the arrow indicates a 1.75 eV peak existing in the FeSe$_2$ sample.
IV. SUMMARY AND CONCLUSIONS

FeSe thin films with different $T_{c0}$ fabricated by the PLD technique exhibit a high-quality single crystal measured by XRD and the lattice constant is about 5.574 Å. The surface topography characterized by AFM shows that light stripes exist in FeSe thin films. It is helpful to construct an optical structure model in SE data analysis and offer the thickness of roughness. Using the Drude and Lorentz dispersion laws to fit $\tan\Psi$ and $\cos\Delta$, complex refractive index or dielectric functions are derived from calculating scattering rate $\tau^{-1}$. The result turns out that $\tau^{-1}$ of FeSe$_2$ with 11 K transition temperature is significantly lower than that of FeSe$_1$ with $T_{c0} = 6$ K at room temperature, which may be the reason that FeSe$_2$ exhibits higher $T_{c0}$ in low temperature than FeSe$_1$. Therefore, we provide a feasible way to relate optical constants with electric quantities for the FeSe system.

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